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# Synergistically boosting the elementary reactions over multiheterogeneous ordered macroporous Mo<sub>2</sub>C/NC-Ru for highly efficient alkaline hydrogen evolution

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### Abstract

Simultaneously enhancing the reaction kinetics, mass transport, and gas release during alkaline hydrogen evolution reaction (HER) is critical to minimizing the reaction polarization resistance, but remains a big challenge. Through rational design of a hierarchical multiheterogeneous three-dimensionally (3D) ordered macroporous Mo<sub>2</sub>C-embedded nitrogen-doped carbon with ultrafine Ru nanoclusters anchored on its surface (OMS Mo<sub>2</sub>C/NC-Ru), we realize both electronic and morphologic engineering of the catalyst to maximize the electrocatalysis performance. The formed Ru-NC heterostructure shows regulative electronic states and optimized adsorption energy with the intermediate H\*, and the Mo<sub>2</sub>C-NC heterostructure accelerates the Volmer reaction due to the strong water dissociation ability as confirmed by theoretical calculations. Consequently, superior HER activity in alkaline solution with an extremely low overpotential of 15.5 mV at  $10 \text{ mA cm}^{-2}$  with the mass activity more than 17 times higher than that of the benchmark Pt/C, an ultrasmall Tafel slope of  $22.7 \text{ mV dec}^{-1}$ , and excellent electrocatalytic durability were achieved, attributing to the enhanced mass transport and favorable gas release process endowed from the unique OMS Mo<sub>2</sub>C/NC-Ru structure. By oxidizing OMS Mo<sub>2</sub>C/NC-Ru into OMS MoO<sub>3</sub>-RuO<sub>2</sub> catalyst, it can also be applied as efficient oxygen evolution electrocatalyst, enabling the

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#### **KEYWORDS**

heterostructure, hydrogen evolution reaction, molybdenum carbide, ordered macroporous structure, ruthenium nanoparticle, synergistic effect

### **1** | INTRODUCTION

Climate change due to the excessive use of fossil fuels has been pushing people to explore new carbon-neutral renewable energy sources.<sup>1,2</sup> Hydrogen, as one of the most promising alternatives, has attracted great attention owing to its high energy density, sustainability, and cleanliness.<sup>3–5</sup> Water electrolysis using renewable energy input (such as solar and wind energy) provides an attractive and eco-friendly approach for large-scale hydrogen production.<sup>6-8</sup> Compared with the acidic water splitting technique, the alkaline water splitting in anion-exchangemembrane water electrolyzers (AEMWEs) can reduce the damage of electrolyte to metal-based electrode materials and increase the choices of available electrocatalysts due to the milder alkaline working environment.<sup>9</sup> Therefore, the research on AEMWEs has become a hot topic in the field of energy conversion. Till now, even in alkaline solution, platinum (Pt)-based materials are still the benchmark electrocatalysts for electrochemical hydrogen evolution reaction (HER) due to their high activity. However, the Earth's scarcity and high price severely hinder their widespread application.<sup>10,11</sup> To reduce the cost, it is highly imperative to design Pt-free HER catalysts with Pt-comparable activity in an alkaline medium.

As a much cheaper noble metal in the Pt group (only one-third the price of Pt), ruthenium (Ru) exhibits a similar bond strength with hydrogen (ca. 65 kcal mol<sup>-1</sup>) to that of Pt and a favorable chemical inertness to corrosive electrolyte, making it a promising substitute to Pt.<sup>12–14</sup> Although Ru possesses potentially high electrocatalytic activity for HER, the active Ru nanoparticles (NPs) tend to agglomerate because of the large cohesive energy, resulting in decreased activity.<sup>15,16</sup> To resolve this issue, some conductive substrates, especially carbon materials (carbon nanotube,<sup>16</sup> graphene,<sup>17,18</sup> carbon nanofiber,<sup>19</sup> carbon dot,<sup>20,21</sup> C<sub>2</sub>N,<sup>22</sup> various precursor-derived carbon,<sup>23–25</sup> etc.) were often utilized to anchor Ru NPs due to their high conductivity, good stability, and rich functional groups. The carbon support not only provides fast electron transfer but also forms a

heterostructure with Ru NPs, which could regulate the local electronic structures and hydrogen adsorption free energy  $(\Delta G_{\rm H^*})$  of the active sites due to the strong metal-support interaction, thereby modifying the electrocatalytic activity and stability.<sup>18,22,23</sup> However,  $\Delta G_{H^*}$  alone is hardly sufficient to describe the activity of HER in alkaline solutions<sup>26</sup> because of its multistep process in nature. Actually, the Volmer step  $(H_2O + * + e^- \rightarrow H^* + OH^-)$  and the Heyrovsky step  $(H^* + H_2O + e^- \rightarrow H_2 + * + OH^-)$  or the Tafel step  $(H^* + H_2O + e^- \rightarrow H_2 + * + OH^-)$  $H^+ + e^- \rightarrow H_2 + *$ ) act together to affect the overall HER activity.<sup>27,28</sup> In this case, considering the relatively highenergy barrier for water dissociation of Ru,<sup>26</sup> a well-designed substrate with good water dissociation ability is essential to achieving efficient electrocatalysis for alkaline HER. Previous work has confirmed that Mo<sub>2</sub>C possesses a low water dissociation energy in alkali,<sup>29,30</sup> and thus a carbon-Mo<sub>2</sub>C composite support may be a good choice for Ru anchoring.

In addition, to improve the intrinsic activity of electrocatalysts, additional attention should be paid to the mass transport of reactants and hydrogen products during the HER process. While the porous structure could facilitate the mass transfer of reactants, the generated large amount of gas bubbles in the pores might hinder the approach of reactants and shield the catalytically active sites, especially at high current densities, leading to a limited hydrogen production rate.<sup>31,32</sup> Moreover, the bubbles are easily blocked and trapped as they transport through the disordered porous structure, which is not favorable for their rapid release.33,34 Compared to a commonly disordered porous structure containing micropores and mesopores, the well-defined hierarchically porous structure with three-dimensionally (3D) ordered macropores not only guarantees fast mass transfer of reactants but also helps to realize a high gas release rate.<sup>35,36</sup> Taking into account the above views, it is believed that coupling Ru NPs with appropriate carbon-Mo<sub>2</sub>C composite support that possesses hierarchically ordered macroporous structure may achieve superior overall HER performance in alkaline, but this kind of electrocatalyst is rarely explored.

Herein, we reported the rational design of a hierarchically multiheterogeneous OMS Mo<sub>2</sub>C/NC-Ru

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composite by selecting a 3D-ordered macroporous Mo<sub>2</sub>Cembedded nitrogen-doped carbon (OMS Mo<sub>2</sub>C/NC) as support to couple with Ru nanoclusters. The unique structure of OMS Mo<sub>2</sub>C/NC-Ru brought about superior HER performance in 1.0 M KOH electrolyte with an extremely low overpotential of 15.5 mV at a current density of  $10 \text{ mA cm}^{-2}$ , an ultrasmall Tafel slope of 22.7 mV  $dec^{-1}$ , and more than 17 times higher mass activity compared to the benchmark Pt/C electrocatalyst. In addition, after a 100-h long-term cyclic test from 10 to  $100 \text{ mA cm}^{-2}$  using the multicurrent chronopotentiometric method, the overpotential was not changed. Density functional theory (DFT) calculations manifest that the formed Ru-NC heterostructure displayed regulative electronic states and optimized adsorption energy with the intermediate H\*; at the same time, the Mo<sub>2</sub>C-NC heterostructure accelerates the Volmer reaction to offer a much faster proton supply due to the strong water dissociation ability. Thus, the multiheterogeneous synergy greatly boosts the overall HER kinetics in alkaline conditions. After assembling with OMS Mo<sub>2</sub>C/NC-Ruderived ordered macroporous MoO<sub>3</sub>-RuO<sub>2</sub> superstructure (OMS MoO<sub>3</sub>-RuO<sub>2</sub>) electrode, the quasisymmetric electrolyzer requires only a low voltage of 1.62 V to reach 10 mA cm<sup>-2</sup>, much better than that of the state-of-the-art Pt/C || RuO<sub>2</sub> pair.

# 2 | RESULTS AND DISCUSSION

The synthesis process of the OMS Mo<sub>2</sub>C/NC-Ru composite is schematically shown in Scheme 1, which consists of two main steps. First, monodisperse polystyrene spheres (PSs; ~270 nm) were assembled into a 3D-ordered PS template through vacuum filtration (Figure S1), and then an ammonium molybdate tetrahydrate solution was filled into the PS template voids to obtain the  $(NH_4)_6Mo_7O_{24}$ @PS precursor (Figure S2), followed by pyrolysis of the precursor at 750°C under H<sub>2</sub>/Ar atmosphere to decompose PS with the formation of the OMS Mo<sub>2</sub>C/NC. The OMS Mo<sub>2</sub>C/NC-Ru catalyst was then prepared by reducing Ru<sup>3+</sup> with sodium borohydride (NaBH<sub>4</sub>) in the presence of OMS Mo<sub>2</sub>C/NC as the support, followed by subsequent thermal treatment under inert atmospheres.

The scanning electron microscopy (SEM) and transmission electron microscopy (TEM) images of the asprepared OMS Mo<sub>2</sub>C/NC substrate (Figure S3A-C) unveils a 3D framework structure with ordered macroporous cavities (~260 nm) interconnected by macroporous channels with a size of about 110 nm. Figure S3D,E shows that the Mo<sub>2</sub>C nanoparticles were embedded in an NC skeleton with several layers of NC on the surface. The high-angle annular darkfield scanning transmission electron microscopy (HAADF-STEM) image and corresponding elemental mappings of OMS Mo<sub>2</sub>C/NC (Figure S3F) reveals a homogeneous distribution of C, N, and Mo elements in the 3D-ordered macroporous architecture. The X-ray diffraction (XRD) pattern and X-ray photoelectron spectroscopy (XPS) spectrum (Figure S4) further demonstrates the successful formation of the ordered OMS Mo<sub>2</sub>C/NC structure with good crystallinity. In addition, we found that the reduced gas environment and PS template were essential for the formation of OMS Mo<sub>2</sub>C/NC with a pure Mo<sub>2</sub>C phase. Under the N<sub>2</sub> atmosphere, only a mixed phase of molybdenum carbide, oxide, and nitride was obtained (Figure S5). Moreover, in this study, PS was used not only



**SCHEME 1** Schematic illustration of the synthetic process of ordered macroporous Mo<sub>2</sub>C-embedded nitrogen-doped carbon with ultrafine Ru nanoclusters anchored on its surface (OMS Mo<sub>2</sub>C/NC-Ru). PS, polystyrene sphere

as a template but also as a carbon source, despite its complete degradation and volatilization in an inert gas (Figure S6), which may be due to the fact that the generated Mo metal transition phase catalyzed the small molecular hydrocarbons produced by PS decomposition during the pyrolysis (Figure S7), thereby forming a carbon layer deposited on the surface of  $Mo_2C$ .

The multiheterogeneous OMS  $Mo_2C/NC$ -Ru was obtained by simply coupling Ru NPs with the OMS  $Mo_2C/NC$ support. The crystal structure of OMS  $Mo_2C/NC$ -Ru was analyzed by XRD (Figure 1A). All the diffraction peaks can be well indexed based on the hexagonal  $\beta$ -Mo<sub>2</sub>C (PDF#35-0787),<sup>37</sup> whereas no obvious characteristic peaks of Ru NPs were observed, indicating the low content and ultrasmall size of the Ru NPs.<sup>23</sup> The morphology of the catalyst was characterized by SEM and TEM. The SEM images of OMS  $Mo_2C/NC$ -Ru (Figures 1B and S8A,B) demonstrate that the

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ordered macroporous structure was well preserved without significant change in the dimension of macropores after the implantation of Ru. The TEM image also confirms the ordered macroporous framework of OMS Mo2C/NC-Ru (-Figure 1C). Besides, some mesopores were observed on the wall of the macropore (Figure 1D), demonstrating the hierarchically porous structure of the sample, which can effectively enhance the accessibility of the active sites to reactants.<sup>38,39</sup> A magnified TEM image (Figure 1E) clearly discloses a homogeneous distribution of the ultrafine Ru nanoclusters (average diameter  $\sim 2.5 \pm 0.5$  nm) on the ordered macroporous framework. The loading of Ru nanoclusters in the OMS Mo<sub>2</sub>C/NC-Ru sample was estimated to be 1.75 wt% by the inductively coupled plasma-optical emission spectrometer (ICP-OES). The high-resolution TEM (HRTEM) image (Figure 1F) further reveals the coexistence of Ru and Mo<sub>2</sub>C phases. The lattice fringes with



**FIGURE 1** (A) X-ray diffraction pattern, (B) SEM image (inset shows a magnified SEM image), and (C–E) transmission electron microscopy (TEM) images of ordered macroporous Mo<sub>2</sub>C-embedded nitrogen-doped carbon with ultrafine Ru nanoclusters anchored on its surface (OMS Mo<sub>2</sub>C/NC-Ru). (F) High-resolution TEM image of OMS Mo<sub>2</sub>C/NC-Ru. (G) High-angle annular darkfield scanning transmission electron microscopy image and the corresponding energy-dispersive X-ray spectroscopy elemental mappings of OMS Mo<sub>2</sub>C/NC-Ru for C, N, Mo, and Ru elements

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spacings of 0.206 and 0.215 nm can be indexed to the (101) and (002) planes of hexagonal Ru, respectively, and the lattice fringes with a spacing of 0.258 nm are attributed to Mo<sub>2</sub>C. Moreover, the HAADF-STEM images and corresponding element mappings (Figures 1G and S8C,D) further proved the successful formation of uniform ultrafine Ru nanoclusters anchoring on the hierarchically ordered macroporous Mo<sub>2</sub>C/NC superstructure, as evidenced by the even distribution of C, N, Mo, and Ru elements in the whole OMS Mo<sub>2</sub>C/NC-Ru architecture. For comparison, solid Mo<sub>2</sub>C/NC-Ru was also synthesized by the same procedure except that the precursor was replaced by the mixture of disordered PSs powder and ammonium molybdate tetrahydrate. The solid Mo<sub>2</sub>C/NC-Ru showed a solid granular structure with homogeneous elements distribution in the architecture exempt of 3D interconnected ordered macropores (Figures S9 and S10). XRD and XPS results confirm the formation of the solid Mo<sub>2</sub>C/NC-Ru sample (Figure S11).

The pore structure characteristics of OMS  $Mo_2C/NC$ -Ru were investigated by  $N_2$  sorption measurements (Figure 2A). The adsorption/desorption behavior showed a combined II/ IV-type isotherm with H3-type hysteresis loop, confirming the presence of plentiful mesopores,<sup>40</sup> consistent with the TEM observation. OMS Mo<sub>2</sub>C/NC-Ru possessed a larger Brunauer–Emmett–Teller (BET) surface area of 51.7 m<sup>2</sup> g<sup>-1</sup> than OMS Mo<sub>2</sub>C/NC (38.1 m<sup>2</sup> g<sup>-1</sup>) and solid Mo<sub>2</sub>C/NC-Ru (40.0 m<sup>2</sup> g<sup>-1</sup>, Figure S12), which is mainly ascribed to the incorporation of ultrafine Ru nanoclusters in the hierarchically macroporous structure. Raman spectroscopy was used to evaluate the disorder in carbon materials. Similar  $I_D/I_G$  ratio of OMS Mo<sub>2</sub>C/NC-Ru and OMS Mo<sub>2</sub>C/NC indicates that the coupling of Ru nanoclusters had little influence on the structure of N-doped carbon (Figure 2B). A higher  $I_D/I_G$  ratio of OMS Mo<sub>2</sub>C/NC-Ru suggests that it possessed more defects on NC than solid Mo<sub>2</sub>C/NC-Ru (Figure S13), which is favorable for improving the HER performance.<sup>12</sup>

XPS analysis was employed to get insight into the elemental and chemical state of the catalyst surface. The full XPS survey spectrum shows the presence of C, N, Mo, Ru, and O elements in the OMS  $Mo_2C/NC$ -Ru sample (Figure S14). The high-resolution C 1s spectrum can be divided into four peaks with binding energies of 284.7, 285.6, 286.5, and 288.8 eV, corresponding to C-C/C=C, C-N, C-O, and O-C=O, respectively.<sup>25,36</sup>



**FIGURE 2** (A) N<sub>2</sub> sorption isotherms and (B) Raman spectra of OMS Mo<sub>2</sub>C/NC and OMS Mo<sub>2</sub>C/NC-Ru. Inset in (A) shows the corresponding pore size distribution. (C) High-resolution C 1s + Ru 3d and (D) Mo 3d XPS spectra of OMS Mo<sub>2</sub>C/NC-Ru. (E) High-resolution Ru 3p XPS spectrum of OMS Mo<sub>2</sub>C/NC-Ru and Ru NPs. (F) High-resolution N 1s XPS spectra of OMS Mo<sub>2</sub>C/NC-Ru and OMS Mo<sub>2</sub>C/NC-Ru, ordered macroporous Mo<sub>2</sub>C-embedded nitrogen-doped carbon with ultrafine Ru nanoclusters anchored on its surface; XPS, X-ray photoelectron spectroscopy

The partially coincident Ru 3d signal peaks, existing at 281.0 and 284.2 eV, were assigned to the Ru  $3d_{5/2}$  and Ru  $3d_{3/2}$  of the metal Ru, respectively, while the peak at 281.9 eV was related to the oxidized Ru species (Figure 2C).<sup>29,41</sup> In the Mo 3d spectrum, the Mo<sup>4+</sup> and  $Mo^{6+}$  signals were also observed except  $Mo^{2+}$ , attributed to the unavoidable surface oxidation of Mo species when exposed to air (Figure 2D).<sup>42,43</sup> To understand the electronic state of Ru in OMS Mo<sub>2</sub>C/NC-Ru, pure Ru NPs were also prepared for comparison (Figure S15). From the deconvoluted spectrum of Ru 3p (Figure 2E), two peaks at 462.9 and 484.9 eV were attributed to Ru 3p<sub>3/2</sub> and Ru  $3p_{1/2}$  of metallic Ru,<sup>12,21</sup> respectively, while the other two peaks were associated with surface-oxidized Ru species caused by the exposure to air. Besides, the Ru 3p peak of OMS Mo<sub>2</sub>C/NC-Ru showed a positive shift of 0.4 eV compared with that of Ru NPs, suggesting the increased valence state of Ru in OMS Mo<sub>2</sub>C/NC-Ru due to the strong interaction between Ru metal and the Mo<sub>2</sub>C/NC support. At the same time, the N 1s peak of OMS Mo<sub>2</sub>C/NC-Ru displayed a negative shift compared with that of OMS Mo<sub>2</sub>C/NC (Figure 2F), which certified the electron transfer from Ru to N.

The electrocatalytic performance toward HER of OMS Mo<sub>2</sub>C/NC-Ru was investigated in an alkaline solution (1.0 M KOH) using a standard three-electrode system. Pure Ru NPs, OMS Mo<sub>2</sub>C/NC, solid Mo<sub>2</sub>C/NC-Ru, and commercial 20% Pt/C were also measured for comparison. Figure 3A shows the *iR*-corrected HER linear sweep voltammetry (LSV) curves of different catalysts. Pure Ru NPs and OMS Mo<sub>2</sub>C/NC achieved a current density of 10 mA cm<sup>-2</sup> at a relatively high overpotential of 61.9 and 124.1 mV, respectively, while when coupling them together to form a multiheterogeneous OMS Mo<sub>2</sub>C/NC-Ru structure, the overpotential was sharply decreased to 15.5 mV, demonstrating the synergistic effect between Ru NPs and Mo<sub>2</sub>C/NC toward HER catalysis. Notably, OMS Mo<sub>2</sub>C/NC-Ru showed lower overpotential than solid Mo<sub>2</sub>C/NC-Ru (39.3 mV) and Pt/C (23.3 mV). Besides, OMS Mo<sub>2</sub>C/NC-Ru displayed the lowest overpotential of 42.7 mV at 50 mA  $cm^{-2}$ , surpassing all control catalysts (Figure 3B). The Tafel slopes derived from LSV curves were employed to evaluate the HER kinetics. Figure 3C shows that the Tafel slope for OMS  $Mo_2C/NC$ -Ru is 22.7 mV dec<sup>-1</sup>, smaller than that of Pt/C (28.1 mV dec<sup>-1</sup>) and other samples, signifying more efficient hydrogen evolution kinetics and dominant Volmer-Tafel mechanism in the HER pathway.<sup>12,44</sup> From the Tafel plots, the exchange current density of OMS Mo<sub>2</sub>C/NC-Ru is 2.08 mA cm<sup>-2</sup> (Figure S16), which is much larger than those of other catalysts and even higher than that of Pt/C (1.64 mA cm<sup>-2</sup>), indicating its intrinsic rapid HER kinetics.<sup>16,45</sup>

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Impressively, the comparison of activity with other recently reported high-performance HER electrocatalysts demonstrates that our catalyst ranks among the top alkaline HER electrocatalysts (Figure 3D and Table S1). We further evaluated the mass activities of noble metals (Ru and Pt) in OMS Mo<sub>2</sub>C/NC-Ru and Pt/C by normalizing the LSV curves. As shown in Figure 3E, OMS Mo<sub>2</sub>C/NC-Ru possessed extraordinarily high mass activity compared to Pt/C in which, at an overpotential of 50 mV, OMS Mo<sub>2</sub>C/NC-Ru showed the mass activity of 1.118 A mg<sup>-1</sup><sub>Ru</sub>, 17.8 times higher than that of Pt/C (0.0627 A mg<sup>-1</sup><sub>Pt</sub>). The high mass activity of the catalyst is undoubtedly conducive to its low-cost practical applications.

As an important parameter, the electrochemical surface area (ECSA) for HER was estimated by the electrochemical double-layer capacitance  $(C_{dl})$  in a non-Faradaic region (Figure S17). As shown in Figure 3F, OMS Mo<sub>2</sub>C/ NC-Ru exhibited the  $C_{dl}$  of 15.5 mF cm<sup>-2</sup>, which is higher than those of OMS  $Mo_2C/NC$  (3.5 mF cm<sup>-2</sup>), Ru NPs  $(4.8 \text{ mF cm}^{-2})$ , and Solid Mo<sub>2</sub>C/NC-Ru  $(9.9 \text{ mF cm}^{-2})$ , indicating that coupling Ru nanoclusters with the hierarchically ordered macroporous Mo<sub>2</sub>C/NC is beneficial to exposing more active sites, thus enhancing the catalytic activities.<sup>36</sup> Electrochemical impedance spectroscopy was further performed to study the interfacial electron transfer dynamics during the HER process. OMS Mo<sub>2</sub>C/NC-Ru demonstrated the smallest charge transfer resistance ( $R_{ct}$ ) of 1.75  $\Omega$  (Figures 3G and S18), suggesting the fastest charge transfer at the interface of OMS Mo<sub>2</sub>C/NC-Ru and the electrolyte, which improved the HER performance. The dynamic water contact angle measurement confirmed the superior hydrophilic nature of OMS Mo<sub>2</sub>C/NC-Ru (Figure S19), indicating that the hierarchically ordered macroporous architecture facilitated the penetration of the electrolyte and enhanced the mass transport.<sup>25,46-48</sup> Notably, such a unique OMS Mo<sub>2</sub>C/NC-Ru structure endowed the superior hydrophilicity, which is favorable for activating water molecules and accelerating the Volmer step in the HER process.<sup>28</sup> To get more insights into the active sites of OMS Mo<sub>2</sub>C/NC-Ru, thiocyanate ions (SCN<sup>-</sup>) were added to the KOH electrolyte as metal-centered active site poisoners (Figure S20). It was found that the current density of OMS Mo<sub>2</sub>C/NC-Ru decreased dramatically with the loss ratio of 85.1% upon the addition of SCN<sup>-</sup>, indicating that the active sites were blocked by SCN<sup>-</sup>. As a comparison, the current density of OMS Mo<sub>2</sub>C/NC decreased relatively slowly, with the loss ratio of 28.4%, which unveils Ru nanoclusters as the major active sites of the OMS Mo<sub>2</sub>C/NC-Ru catalyst.

The electrocatalytic durability of OMS Mo<sub>2</sub>C/NC-Ru was examined to evaluate the HER performance.



FIGURE 3 Electrocatalytic HER performance evaluation of various catalysts in 1.0 M KOH. (A) HER polarization curves of Pt/C, OMS Mo<sub>2</sub>C/NC-Ru, solid Mo<sub>2</sub>C/NC-Ru, OMS Mo<sub>2</sub>C/NC, and Ru NPs catalysts with iR correction. (B) Comparison of overpotential changes at 10, 20, and 50 mA cm<sup>-2</sup>. (C) Tafel plots obtained from the polarization curves in (A). (D) Comparison of the Tafel slope and overpotential at 10 mA cm<sup>-2</sup> in 1.0 M KOH for OMS Mo<sub>2</sub>C/NC-Ru with other recently reported HER catalysts. (E) Mass activities of OMS Mo<sub>2</sub>C/NC-Ru and Pt/C. (F) The differences in current density ( $\Delta j$ ) at 0.975 V (vs. RHE) as a function of scan rate for OMS Mo<sub>2</sub>C/NC-Ru, solid Mo<sub>2</sub>C/NC-Ru, OMS  $Mo_2C/NC$ , and Ru NPs. Inset is the corresponding  $C_{dl}$  values. (G) Electrochemical impedance spectroscopy of Pt/C, OMS  $Mo_2C/NC$ -Ru, solid Mo<sub>2</sub>C/NC-Ru, OMS Mo<sub>2</sub>C/NC, and Ru NPs. Inset is the equivalent circuit in which R<sub>s</sub>, R<sub>ct</sub>, and CPE are solution resistance, charge transfer resistance between catalyst and electrolyte, and constant phase element, respectively. (H) Multicurrent chronopotentiometric curves of OMS Mo<sub>2</sub>C/NC-Ru and Pt/C with iR correction. HER, hydrogen evolution reaction; NP, nanoparticle; OMS Mo<sub>2</sub>C/NC-Ru, ordered macroporous Mo<sub>2</sub>C-embedded nitrogen-doped carbon with ultrafine Ru nanoclusters anchored on its surface

The LSV curves almost remained unchanged and the overpotential showed an inappreciable decline before and after 5000 cyclic voltammetry (CV) cycles (-Figure S21), demonstrating good long-term stability, which was also verified by the *i*-t chronoamperometric response of OMS Mo<sub>2</sub>C/NC-Ru over 40 h. Additionally, the multicurrent chronopotentiometric measurement was carried out to further assess the outstanding stability of OMS Mo<sub>2</sub>C/NC-Ru with Pt/C as a contrast

(Figure 3H). It can be seen that OMS Mo<sub>2</sub>C/NC-Ru exhibited better stability throughout the 100 h cyclic test at different current densities from 10 to  $100 \text{ mA cm}^{-2}$  with smaller overpotential increments as the current density increases. Significantly, the curve of Pt/C is rougher than that of OMS Mo<sub>2</sub>C/NC-Ru, especially at high current densities of 100 mA cm<sup>-2</sup>, suggesting a greater disturbance to the Pt/C catalyst by plenty of gas bubbles in the HER process, which may be ascribed to the lack of hierarchically ordered macropores for fast gas release. Meanwhile, we studied the change of the structure and composition of OMS  $Mo_2C/NC$ -Ru before and after HER. The ordered macroporous structure was well maintained after the stability test (Figures S22 and S23) and the composition was basically unchanged (Figure S24). These results indicate the superior chemical and structural stability of OMS  $Mo_2C/NC$ -Ru catalyst during the HER process, which are the cornerstone of the excellent electrocatalytic performance.

To elucidate the underlying mechanism of superior HER activity of OMS Mo<sub>2</sub>C/NC-Ru, DFT calculations were conducted. The theoretical models of Ru, Ru-NC,

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and Mo<sub>2</sub>C-NC heterointerfaces were established based on the experimental observation and analysis (Figures 4A and S25). In alkaline media, favorable H<sub>2</sub>O adsorption on catalysts is important for subsequent effective H<sub>2</sub>O activation.<sup>49</sup> Compared with Ru-NC, Mo<sub>2</sub>C-NC has a lower H<sub>2</sub>O adsorption energy value of -0.64 eV (Figure 4B), leading to much faster H<sub>2</sub>O adsorption on the surface of Mo<sub>2</sub>C-NC, which is beneficial to activating water molecules and accelerating the first step of Volmer reaction.<sup>22,28</sup> In addition, the rapid dissociation of adsorbed H<sub>2</sub>O into H<sup>\*</sup> and \*OH guarantees sufficient protons for the overall reaction. As shown in Figure 4C, the dissociation of H<sub>2</sub>O on the surface of Mo<sub>2</sub>C-NC is much easier



**FIGURE 4** (A) Structures of Ru-NC and Mo<sub>2</sub>C-NC. (B) Adsorption energies of  $H_2O$  and  $OH^-$  on Ru, Ru-NC, and Mo<sub>2</sub>C-NC. (C) Gibbs free energy diagram for HER on Ru, Ru-NC, and Mo<sub>2</sub>C-NC. Insets are the geometric configurations of the \* $H_2O$ , H\* + \*OH, and H\* adsorbed on Ru, Ru-NC, and Mo<sub>2</sub>C-NC. (D) Side view of charge density differences in the heterostructures for Ru-NC and Mo<sub>2</sub>C-NC. (E) Density of states (DOS) of Ru, Ru-NC, Mo<sub>2</sub>C, and Mo<sub>2</sub>C-NC systems. (F) Schematic representation for the overall alkaline water splitting. (G) Polarization curves of the OMS Mo<sub>2</sub>C/NC-Ru || OMS MoO<sub>3</sub>-RuO<sub>2</sub> and Pt/C || RuO<sub>2</sub> pairs for water splitting. HER, hydrogen evolution reaction; OMS Mo<sub>2</sub>C/NC-Ru, ordered macroporous Mo<sub>2</sub>C-embedded nitrogen-doped carbon with ultrafine Ru nanoclusters anchored on its surface

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compared with those on the surfaces of Ru-NC and Ru, indicating the faster proton supply by the Mo<sub>2</sub>C-NC carrier. As more \*OH adsorbed on the surface of catalysts may decrease the HER efficiency, we further calculated and compared the adsorption energy of \*OH (Figure 4B). Results show that the formation of Ru-NC heterojunction results in weaker attraction to \*OH compared with pure Ru. Besides, \*OH tends to be adsorbed on the surface of Mo<sub>2</sub>C-NC due to the lower adsorption energy, suggesting that \*OH adsorption on Ru-NC may transfer to that on Mo<sub>2</sub>C-NC, and more active sites in Ru-NC can be released for H\* adsorption, thus promoting the Tafel step. Moreover,  $\Delta G_{H^*}$  is widely used as a critical activity descriptor to evaluate HER performance, and a  $\Delta G_{H^*}$  value close to 0 eV is satisfactory.<sup>50,51</sup> The calculated  $\Delta G_{H^*}$  values of Ru and Mo<sub>2</sub>C-NC are -0.4 and -0.54 eV, respectively, which indicates their strong hydrogen adsorption and unfavorable release process. In contrast, the  $\Delta G_{H^*}$  value at the Ru-NC heterointerface is 0.13 eV, closer to 0, proving its faster hydrogen absorption/desorption kinetics.

To further gain perceptions about the activity origin of OMS Mo<sub>2</sub>C/NC-Ru, the intrinsic electronic states of the formed Ru-NC and Mo<sub>2</sub>C-NC heterointerfaces were calculated. The charge density difference plots reveal that the charge density is redistributed at the constructed heterointerface (Figures 4D and S26) and the electrons transfer from Ru atoms to N atoms, which is consistent with the XPS analysis. The modulated electronic structure is conducive for appropriate adsorption/desorption behavior of reaction intermediates, thus enhancing the catalytic activity. Additionally, in the heterostructures, the density of states (DOS) occupied by the metal d-orbital at the Fermi level is significantly weakened and the overall DOS is more diffuse (Figure 4E), leading to weaker adsorption of the reactants, which is also consistent with the above DFT results. In brief, these theoretical results demonstrate that the Mo<sub>2</sub>C-NC heterojunctions accelerate the water adsorption and dissociation in the Volmer step, and the Ru-NC heterojunctions exhibit an optimized H\* adsorption/ desorption behavior in the Tafel step (Figure 4F), therefore resulting in outstanding HER performance of multiheterogeneous OMS Mo<sub>2</sub>C/NC-Ru in alkaline solution.

Considering the impressive HER performance of OMS  $Mo_2C/NC$ -Ru, it is encouraged to achieve practical overall water splitting by coupling a suitable oxygen evolution reaction (OER) catalyst. Inspired by the good OER activity of commercial RuO<sub>2</sub>, a series of RuO<sub>2</sub>-based catalysts were prepared by annealing OMS  $Mo_2C/NC$ -Ru in the air at different temperatures. The derived materials were denoted as OMS  $MoO_3$ -RuO<sub>2</sub>-X (X represents the temperature) according to the XRD patterns (Figure S27). The OER activities of the as-synthesized OMS  $MoO_3$ -RuO<sub>2</sub>-X were measured using a typical three-electrode system under

1.0 M KOH and compared with that of commercial RuO<sub>2</sub>. Figure S28 displays that the OMS MoO<sub>3</sub>-RuO<sub>2</sub> (derived at 450°C) only needs an overpotential of 380 mV to reach  $10 \text{ mA cm}^{-2}$ , which is 38 mV lower than that of RuO<sub>2</sub> catalyst. The morphology and structure analysis confirmed the formation of OMS MoO<sub>3</sub>-RuO<sub>2</sub> with small RuO<sub>2</sub> NPs loaded on the surface of MoO<sub>3</sub> (Figures S29-S31). Then, we assembled a two-electrode cell using OMS Mo<sub>2</sub>C/NC-Ru as the cathode and OMS MoO<sub>3</sub>-RuO<sub>2</sub> as anode for overall water splitting (Figure 4F). The polarization curves (Figure 4G) show that the OMS Mo<sub>2</sub>C/NC-Ru || OMS MoO<sub>3</sub>-RuO<sub>2</sub> pair required the cell voltages of 1.62 V to drive a current density of  $10 \text{ mA cm}^{-2}$ , which is superior to that of commercial Pt/C || RuO<sub>2</sub> (1.66 V). Besides, Mo<sub>2</sub>C/ NC-Ru || OMS MoO<sub>3</sub>-RuO<sub>2</sub> possessed smaller charge transfer resistance, higher and more stable current density than  $Pt/C \parallel RuO_2$  under the same voltage (Figure S32), suggesting the excellent water splitting performance.

# 3 | CONCLUSION

In summary, we have developed a facile method to construct a hierarchically multiheterogeneous ordered macroporous Mo<sub>2</sub>C/NC-Ru for robust hydrogen production. The formed Mo<sub>2</sub>C-NC and Ru-NC heterointerfaces exhibit a synergistic effect, boosting the Volmer and the Tafel steps in the overall alkaline HER, respectively. Furthermore, the hierarchically ordered macroporous structure endowed the catalyst with enhanced mass transport and a favorable gas release process. As a result, the well-designed OMS Mo<sub>2</sub>C/NC-Ru showed superior HER activity in 1.0 M KOH with an extremely low overpotential of 15.5 mV at  $10 \text{ mA cm}^{-2}$ , an ultrasmall Tafel slope of  $22.7 \text{ mV dec}^{-1}$ , and excellent electrocatalytic durability, outperforming the commercial Pt/C. After assembling with OMS Mo<sub>2</sub>C/NC-Ru-derived OMS MoO<sub>3</sub>-RuO<sub>2</sub> to form a practical water-splitting cell, the performance also surpassed that of the state-of-the-art Pt/ C || RuO<sub>2</sub> electrolyzer. This study provides instructive guidance for designing 3D-ordered macroporous multicomponent catalysts for efficient catalytic application.

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# **CONFLICTS OF INTEREST**

The authors declare no conflicts of interest.

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